

Magnesium diiodide,  $\text{MgI}_2$ Michael A. Brogan, Alexander J. Blake, Claire Wilson and  
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Single crystals of magnesium diiodide have been grown and the structure solved for the first time from single-crystal X-ray diffraction data. This study confirms that  $\text{MgI}_2$  is isostructural with  $\text{CdI}_2$  (C6 or 2H structure type). The space group is  $P\bar{3}m1$  with the Mg atom on a site with  $3m$  symmetry (Wyckoff site 1a) and the I atom on a site with  $m$  symmetry (Wyckoff site 2d). Trends in the 2H structures of dihalides are discussed briefly.

## Comment

$\text{MgI}_2$  belongs to the series of known dihalides ( $\text{AX}_2$ , where  $X = \text{F}, \text{Cl}, \text{Br}, \text{I}$ ). Surprisingly, single-crystal growth of  $\text{MgI}_2$  has never been reported and no refined structure exists. Seventy years ago, Blum (1933) derived a structural model from powder X-ray diffraction data. The conclusion reached was that  $\text{MgI}_2$  crystallizes in a layered structure isotopic with  $\text{CdI}_2$  (defined as C6 or 2H type). Here, the growth of good quality single crystals has allowed the structure of  $\text{MgI}_2$  to be determined for the first time.

The single-crystal diffraction data set obtained confirms that  $\text{MgI}_2$  crystallizes in the trigonal space group  $P\bar{3}m1$ , adopting the same hexagonal close-packed layered structure as  $\text{CdI}_2$ , with unit-cell parameters  $a = 4.1537(7) \text{ \AA}$ ,  $c = 6.862(2) \text{ \AA}$ ,  $V = 102.53(4) \text{ \AA}^3$ ,  $Z = 1$ , and  $c/a = 1.652$ . Within the structure, the  $\text{I}^-$  anions are hexagonally close-packed, whilst the  $\text{Mg}^{2+}$  cations occupy all of the octahedral holes between alternate layers of close-packed anions. Hence, half of the available octahedral holes throughout the structure are occupied. The structure can also be thought of as being built up from layers of  $\text{MgI}_6$  octahedra. Each such octahedron shares edges with six adjacent identical units within each  $\text{MgI}_2$  layer (Fig. 1).

The six equal Mg–I bond distances of  $2.9183(5) \text{ \AA}$  and the I–Mg–I bond angles ( $\phi$ ) of  $90.739(17)^\circ$  describe a slightly distorted  $\text{MgI}_6$  octahedron. Interestingly, the only other example of a magnesium iodide in the literature with a refined

crystal structure is the ternary compound  $\text{CsMgI}_3$  (McPherson *et al.*, 1980), in which the Mg–I distance is  $2.899(4) \text{ \AA}$ , in good agreement with the value obtained in the present work. Apart from the original work of Blum (1933), the only other measurement of the Mg–I distance in  $\text{MgI}_2$  was performed by Akishin & Spiridonov (1958), who used electron diffraction to determine the bond length in the gas phase. Unsurprisingly, the value of  $2.52 \text{ \AA}$  in the triatomic structure is significantly shorter than the value observed here in the solid state. The  $M-X$  distances differ slightly from the corresponding values in  $\text{CdI}_2$  ( $2.989 \text{ \AA}$ ; Pałosz & Salje, 1989), in which the octahedral  $\text{CdI}_6$  unit is slightly more distorted than the  $\text{MgI}_6$  octahedron, with I–Cd–I bond angles of  $90.49^\circ$ . Iodide anions in  $\text{MgI}_2$  are separated by intraplanar distances of  $4.1009(11)$  and  $4.1537(7) \text{ \AA}$  across the thickness and along the length of the Mg–I layer, respectively. Magnesium cations are similarly separated within layers by  $4.1537(7) \text{ \AA}$ . Iodide ions in adjacent layers are separated by a longer interplanar distance of  $4.2722(12) \text{ \AA}$ .

The notable feature of the present diiodide and the structure type in general is the van der Waals interlayer gap between hexagonally close-packed  $\text{I}^-$  ions. The interlayer gap ( $d$ ) in  $\text{MgI}_2$  [ $3.535(2) \text{ \AA}$ ] is significantly wider than the Mg–I layer thickness ( $t$ ) [ $3.326(2) \text{ \AA}$ ]. The same trend is observed in  $\text{CdI}_2$ , but the difference between these two distances is far smaller. Furthermore, by contrast, the absolute value of  $d$  in  $\text{CdI}_2$  is smaller than in  $\text{MgI}_2$ , whereas  $t$  is larger. The relevant structural information for both  $\text{MgI}_2$  and  $\text{CdI}_2$  appears in Table 2. The values for other isostructural dihalides are included for comparison. Note that the data for  $\text{FeCl}_2$  represent the high-pressure polymorph at  $6.4 \text{ Kbar}$  ( $1 \text{ bar} = 10^5 \text{ Pa}$ ) (Vettier & Yelon, 1975).

The  $c/a$  ratio observed here for  $\text{MgI}_2$  is in close agreement with that previously reported (1.66; Blum, 1933). The  $t/d$  ratio obtained previously for  $\text{MgI}_2$  can only be inferred as 1 assuming the  $z$  position for I is  $\frac{1}{4}$  as reported. Table 2 reveals that the vast majority of dihalides with the  $\text{CdI}_2$  structure have  $c/a$  values in the range  $1.6 \leq c/a \leq 1.65$ .  $\text{PbI}_2$  is perhaps the

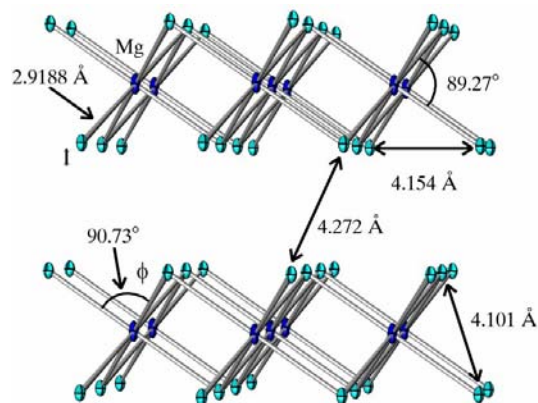
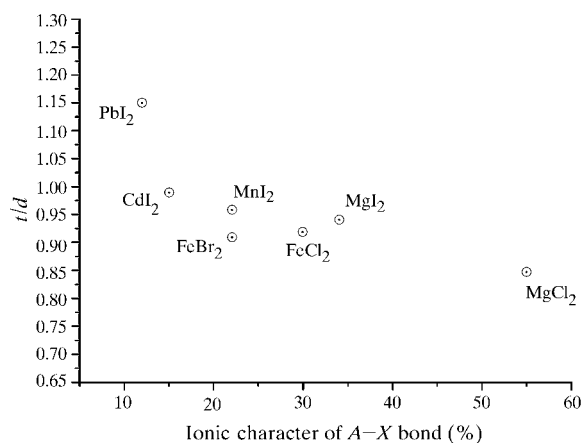


Figure 1

The coordination environment of  $\text{Mg}^{2+}$  (dark ellipsoids) and  $\text{I}^-$  (light ellipsoids) within octahedral layers. Displacement ellipsoids are drawn at the 50% probability level.



**Figure 2**  
Plot of the layer thickness/interlayer gap ratio ( $t/d$ ) versus ionic character (ionicity) of the  $A-X$  bond. Data are taken from Table 2.

most notable exception in this respect, in that it has the largest  $a$  parameter and  $A-X$  bond length, and consequently the smallest  $ca$  ratio reported (Pałosz *et al.*, 1990). Interestingly,  $PbI_2$  is the only dihalide with  $t/d > 1$ . In fact, only a select number of the dihalide 2H structures reported in the literature and depicted in Table 2 have been refined. In the remainder of cases, the halide  $z$  coordinate,  $z(X)$ , is assumed to be 0.25. A number of trends are nonetheless evident from the pool of refined data. As might be expected, the  $t/d$  ratio increases with increasing  $A-X$  bond length and decreasing  $X-A-X$  bond angle ( $\varphi$ ). Bonding in the essentially ionic layers (manifested in  $t$ ) generally plays a more significant role here than the weaker anion repulsion between layers (influencing  $d$ ). Naturally, the halide  $z$  coordinate ( $2d$  site;  $\frac{1}{3}, \frac{2}{3}, z$ ) follows the same trends with  $A-X$  and  $\varphi$ . Interestingly, again here,  $PbI_2$  also possesses the most covalent  $A-X$  bond, thus minimizing interlayer repulsion. It is, therefore, not so unexpected that  $d < t$  in this compound. In fact, there is a good correlation between the ionic character of the  $A-X$  bond, as defined by Pauling (1960), and the  $t/d$  ratio (Fig. 2). A similar correspondence obviously exists between  $z(X)$  and the bond ionicity. The  $ca$  ratio is broadly constant across the structure type (and invariant with  $A-X$  and  $\varphi$ ) but, within the observed range, increases slightly with decreasing  $t/d$  [ $z(X)$ ].

As the number of accurately refined metal dihalide structures increases, the relationships between the various structural variables should become clearer to the point where it should become possible to predict (interpolate)  $z(X)$ , and hence bond length, on the basis of the  $A-X$  bond ionicity or the  $ca$  ratio. The continuing advances made in structure determination from powder data should facilitate and expediate this process.

## Experimental

The title  $MgI_2$  crystals were produced during the attempted reaction of  $MgI_2$  (Aldrich, anhydrous beads 99.9%, ground into a poly-

crystalline powder) with  $Mg_3N_2$ .  $Mg_3N_2$  powder was prepared by the reaction of pure magnesium (Strem Chemicals, 99.9% purity) with dry nitrogen gas at 973 K. Due to the air sensitivity of the reactants and products involved, all manipulations were carried out in glove-boxes (either recirculating nitrogen-filled or evacuable argon-filled). Stoichiometric ratios of the reactants were thoroughly mixed and ground together, then pressed to form a pellet (*ca* 1 g). The pellet was placed in a molybdenum foil liner and transferred to a stainless steel crucible, which was subsequently welded shut under an argon atmosphere. The sealed crucibles were heated in a tube furnace (1323 K, 5 d) under flowing argon to prevent oxidation of the steel crucibles. The furnace was cooled slowly (20 K  $h^{-1}$ ). The crucibles were opened in an  $N_2$ -filled glove-box. Opaque irregular crystals were observed on the molybdenum foil. Crystals were selected in a recirculating  $N_2$ -filled glove-box under an optical microscope and placed under RS3000 perfluoropolyether (Riedel de Haën) on a microscope slide prior to mounting on the diffractometer. The moisture-free viscous perfluoropolyether protects the crystals from atmospheric oxygen and moisture without interfering with the diffraction experiment.

## Crystal data

$MgI_2$	Mo $K\alpha$ radiation
$M_r = 278.11$	Cell parameters from 529 reflections
Trigonal, $P\bar{3}m1$	$\theta = 3.0-27.3^\circ$
$a = 4.1537$ (7) Å	$\mu = 15.24$ mm <sup>-1</sup>
$c = 6.862$ (2) Å	$T = 150$ (2) K
$V = 102.53$ (4) Å <sup>3</sup>	Plate, colourless
$Z = 1$	$0.20 \times 0.20 \times 0.08$ mm
$D_x = 4.504$ Mg m <sup>-3</sup>	

**Table 1**

Selected geometric parameters (Å, °).

Mg—I	2.9183 (5)
Mg <sup>i</sup> —I—Mg	90.739 (17)

Symmetry code: (i)  $1+x, y, z$ .

**Table 2**

Selected structural data (Å, °) for  $MX_2$  dihalides with the  $CdI_2$  (2H) structure.

$MX_2$	$M-X^a$	$\varphi^a$	$ca$	$t$	$d$	$t/d^b$	$z(X)$
$CdI_2^i$	2.989	90.49	1.62	3.4211	3.443	0.99	0.2492
$PbI_2^{ii}$	3.227	90.16	1.53	3.738	3.248	1.15	0.2675 (2)
$MgI_2$	2.9183 (5)	90.739 (17)	1.65	3.3267	3.5353	0.94	0.24237 (6)
$MnI_2^{iii}$	2.92	90.44	1.65	3.346	3.482	0.96	0.245 (2)
$MnBr_2^{iv}$	2.729	90.27	1.62	3.136	3.136	1.00	0.25
$TiBr_2^v$	2.65	93.59	1.79	3.246	3.246	1.00	0.25
$FeBr_2^{vi}$	2.636	91.47	1.65	2.965	3.261	0.91	0.238 (5)
$VCl_2^{vii}$	2.538	90.32	1.62	2.915	2.915	1.00	0.25
$TiCl_2^{viii}$	2.527	90.40	1.65	2.937	2.937	1.00	0.25
$MgCl_2^{ix}$	2.505	93.21	1.63	2.726	3.2	0.85	0.23
$TiCl_2^x$	2.499	93.35	1.78	3.05	3.05	1.00	0.25
$FeCl_2^{xi}$	2.483	92.41	1.6	2.744	2.99	0.92	0.2393 (2)
$CaI_2^{xii}$	3.12	93.313	1.55	3.48	3.48	1	0.25

† Notes: (a) in some cases, s.u. values are not reported in the literature; (b) where  $t/d = 1$ , the literature assumes that  $z(X) = \frac{1}{4}$ , where  $t$  = thickness of layer and  $d$  = interlayer gap. References: (i) Pałosz & Salje (1989); (ii) Pałosz *et al.* (1990); (iii) Cable *et al.* (1962); (iv) Wollan *et al.* (1958); (v) Ehrlich *et al.* (1961); (vi) Haberecht *et al.* (2001); (vii) Ehrlich & Seifert (1959); (viii) Baenziger & Rundle (1948); (ix) Bassi *et al.* (1982); (x) Gal'perin & Sandler (1962); (xi) Vettier & Yelon (1975); (xii) Blum (1933).

## Data collection

Bruker SMART1000 CCD area-detector diffractometer  
 $\omega$  scans  
 Absorption correction: by integration (*SHELXTL*; Bruker, 2001)  
 $T_{\min} = 0.093$ ,  $T_{\max} = 0.505$   
 623 measured reflections  
 115 independent reflections

112 reflections with  $I > 2\sigma(I)$   
 $R_{\text{int}} = 0.081$   
 $\theta_{\text{max}} = 27.3^\circ$   
 $h = -5 \rightarrow 4$   
 $k = -5 \rightarrow 4$   
 $l = -8 \rightarrow 5$

## Refinement

Refinement on  $F^2$   
 $R[F^2 > 2\sigma(F^2)] = 0.021$   
 $wR(F^2) = 0.063$   
 $S = 1.13$   
 115 reflections  
 7 parameters

$w = 1/[\sigma^2(F_o^2) + (0.037)^2 + 0.102P]$   
 where  $P = (F_o^2 + 2F_c^2)/3$   
 $(\Delta/\sigma)_{\text{max}} = 0.001$   
 $\Delta\rho_{\text{max}} = 0.86 \text{ e } \text{\AA}^{-3}$   
 $\Delta\rho_{\text{min}} = -0.91 \text{ e } \text{\AA}^{-3}$

Data collection: *SMART* (Bruker, 2001); cell refinement: *SAINT* (Bruker, 2002); data reduction: *SAINT* and *SHELXTL* (Bruker, 2001); program(s) used to solve structure: *SIR97* (Altomare *et al.*, 1999); program(s) used to refine structure: *SHELXL97* (Sheldrick, 1997); molecular graphics: *ATOMS* (Dowty, 1998); software used to prepare material for publication: *enCIFer* (CCDC, 2003), *PLATON* (Spek, 1990, 2002) and *WinGX* (Farrugia, 1998).

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Supplementary data for this paper are available from the IUCr electronic archives (Reference: BC1028). Services for accessing these data are described at the back of the journal.

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